

Effect of Humidity on Thermal and Electrical Behaviour of Polyindole/Tungsten Carbide Nanocomposites

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Abstract

Nanocomposites (NC) derived from electrically conductive polymers have emerged as promising materials for advanced applications in sensors, semiconductors and supercapacitors. The performance of these materials is critically influenced by environmental factors, with humidity exposure (HE) playing a pivotal role in determining their thermal and electrical behaviours. PIN, recognized for its high redox activity, tunable conductivity and thermal stability, has been integrated with WC, a material known for its exceptional hardness, wear resistance, high electrical conductivity (σ_{DC}) and thermal resilience. In this study, a novel series of NC electro polymers with WC were synthesized via FeCl₃-initiated chemical oxidative polymerization of indole in CTAB presence. WC was incorporated at varying wt% (5, 10 and 15), to assess its impact on the composites' properties. Influence of relative humidity (40%) on thermal stability and σ_{DC} of NC was systematically evaluated. Structural and morphological analyses were performed to elucidate composites' conductivity, stability and reliability under fluctuating humidity conditions. Results revealed that inclusion of WC significantly enhanced thermal and electrical properties of PIN, while providing superior resistance to humidity-induced degradation. Notably, NC with 15 wt% WC exhibited highest σ_{DC} , achieving 36.4mS/cm after 6h of HE. These findings highlight NC's potential as robust materials for diverse applications in industrial, environmental, medical and agricultural domains, where stability under variable humidity conditions is paramount.

Keywords: band gap; conductivity; HE; nanocomposite; PIN; WC.

Introduction*

In recent decades, electrical conducting polymers (ECP), like PPy, PANi, PIN and their derivatives, have earned significant interest, due to their extensive applications in fields such as sensors, biosensors, solid-state devices, batteries and energy storage systems [1, 2]. Among these ECP, PIN has emerged as a

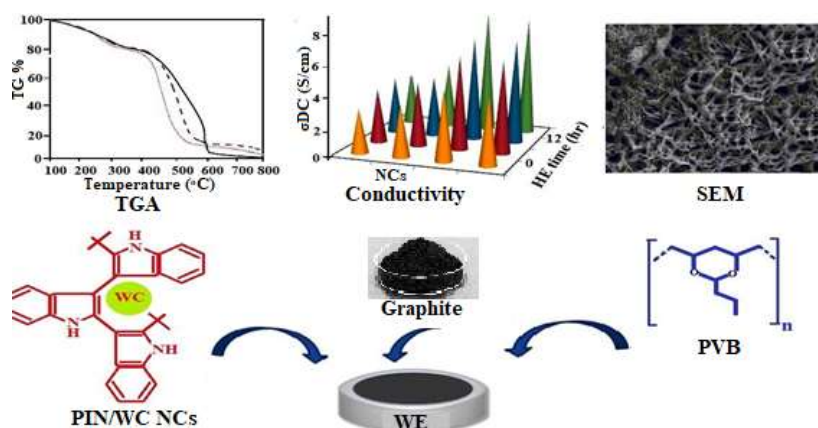
*The abbreviations list is in pages 214-215.

promising polymer for electrical energy storage, owing to its unique combination of properties, including high redox activity, tunable conductivity, thermal stability and a relatively slow degradation rate [3,4]. Despite its advantages, PIN's structural characteristics, such as its non-planar configuration, lightweight nature, loosely packed structure and randomly oriented polymer chains, lead to weak interchain interactions and limited conductivity [5]. Additionally, electrical and electrochemical conductivity of PIN, approximately half that of PPy and PANi [6], has restricted its exploration compared to other ECP. PIN also has challenges such as instability, mechanical strain and capacity loss over extended usage [7]. Addressing these intrinsic limitations is critical to realizing PIN's full potential in practical applications. Recent efforts to overcome these drawbacks include polymeric scaffolding, which enhances structural regularity, minimizes heterogeneity and improves properties such as stereoregularity, electrostatic orientation and doping efficiency, thereby increasing conductivity [8]. Furthermore, integrating PIN with complementary materials as fillers, or within matrices, has been shown to significantly enhance its performance, making it a more viable and efficient material for energy storage applications [9-12].

Metal oxides have been widely employed as inorganic fillers in the fabrication of nanocomposites (NC) derived from PIN. A few examples have been reported such as V_2O_5 , SnO_2 , ZnO , Fe_2O_3 , TiO_2 , NiO , CuO and MnO [9-16]. NC from this category demonstrates significant potential for applications such as corrosion protection [17], sensor development [18] and electrode materials for supercapacitors, due to their superior electrical and electrochemical characteristics. Despite these advancements, limited studies have focused on synthesis and characterization of NC derived from ECP incorporating metal carbides [19-21]. Transition-metal carbides (TMC) are emerging as promising alternatives to noble metals, owing to their cost-effectiveness, exceptional catalytic activity, high selectivity in catalysing specific reactions [22] and superior thermal stability [23]. TMC exhibit a unique combination of physical properties characteristic of covalent solids, ionic crystals and transition metals. Their high electronic conductivity and chemical stability make them particularly valuable for applications in photo- and electrocatalysis [24, 25], as well as energy storage and conversion [26- 28]. Among TMC, tungsten carbide (WC) stands out for its exceptional hardness, wear resistance and toughness, making it indispensable in industries for cutting tools, moulds and dies [29]. WC is recognized as the most metallic carbide, with the highest electrical conductivity (σ_{DC}) (105 S/cm^{-1} at 20°C) among interstitial carbides [30, 31]. Additionally, it is approximately twice as stiff and dense as steel, exhibiting very high thermal conductivity and stability. These unique electrical and physical properties make WC a highly suitable choice for NC preparation [32]. NC's chemical structure and morphology play a crucial role in determining electrode performance, including conductivity, stability and reliability, which are significantly influenced by environmental factors. Among these, HE is a key parameter

affecting NC's performance across a wide range of applications, including industrial, environmental, domestic and medical sectors.

This study involves the synthesis of thermally stable PIN/WC-based NC with enhanced σ_{DC} . WC, known for its cost-effectiveness compared to Pt-based catalysts, and exceptional resistance to thermo-oxidation, serves as ideal filler for producing high-performance NC with superior thermal and electrical properties [30]. This research evaluates thermal stability and σ_{DC} of PIN/WC-based NC, emphasizing the influence of HE on WE. Humidity plays a pivotal role in the reliability and lifespan of electronic devices, often accelerating degradation, and increasing operational costs in unstable environments. Understanding and mitigating these effects is crucial for designing robust and cost-effective materials. Findings highlight the potential of synthesized PIN/WC-based NC to maintain reliable performance under humid conditions, positioning them as durable and versatile materials for diverse practical applications. Scheme 1 shows the steps for producing PIN/WC-based NC.



Scheme 1: Steps involved in the production of PIN/WC-based NC.

Experimental

Materials

PVB (Himedia, Delhi, India), indole (>99%, Himedia, India), THF (>99%, Merck, Mumbai, India), were purchased from above-mentioned firms. Graphite (98.0%, 500 mL) and FeCl_3 (>99%) were acquired from Loba-Chemi, Mumbai, India. CTAB (>99%) and WC (60 nm, >99%) were bought from Sigma Aldrich, Bangalore, India. HCl and acetone (>98%) were procured from Sd. Fine, Chandigarh, India. All chemicals and solvents were of analytical grade, obtained locally, and used without additional purification.

Methodology

Synthesis of PIN

PIN was synthesized via chemical oxidative polymerization in an aqueous medium. Indole (1 g) was dissolved in 30 mL deionized water, and stabilized with CTAB (Fig. 1). The solution was stirred for 6 h, to achieve a uniform

dispersion. Subsequently, an aqueous solution of 3 g FeCl₃ in 50 mL deionized water, was introduced dropwise at a controlled rate of 1 mL/min, under constant stirring. The reaction mixture was maintained under continuous stirring for 24 h, resulting in PIN's precipitation. The precipitate was collected by filtration, washed several times with deionized water, to remove residual impurities, including CTAB [33, 34], which was further dried under reduced pressure (400 mmHg), at 40 °C, for 8 h. The synthesis yielded a blackish-brown polymer, confirming PIN's formation.

Fabrication of PIN-graphite

PVB was initially dissolved in 1.1 mL THF, under continuous magnetic stirring, to ensure solvation. Subsequently, graphite powder was introduced into the PVB solution, and the mixture was magnetically stirred for 2 h, to achieve a uniform dispersion. Following this, PIN was added to the dispersion, and the resultant mixture was stirred for an additional 2 h, to facilitate adequate interaction among constituents. To enhance homogeneity and prevent agglomeration, the mixture underwent ultrasonication, for 30 min. Final PING composite was then uniformly applied onto a stainless steel (SS) plate, to form a consistent coating layer.

Synthesis of PIN/WC-based NC

A suspension of 1 g PIN in 30 mL deionized water was prepared in a flask, and stabilized with 1 g CTAB. The mixture was stirred at 500 rpm, for 6 h, to ensure uniformity. WC suspensions were prepared by dispersing WC powder in an aqueous medium, using ultrasonic agitation to ensure uniform dispersion. CTAB was used to enhance the suspensions' stability, and prevent agglomeration during preparation. WC in varying amounts (5, 10 and 15 wt%) was incorporated into PIN, to produce PIN/WC-based NC, designated as I, II and III, respectively. Following each WC addition, the mixture was ultrasonicated for 30 min. Polymerization was initiated by dropwise addition of a FeCl₃ solution at 1 mL/min, under continuous stirring, for 24 h. Resulting PIN/WC-based NC were filtered, washed with deionized water, to remove unreacted components, and dried under vacuum (400 mm Hg) at 40°C, for 8 h, yielding a brown-coloured product.

Fabrication of working electrode

A series of working electrodes (WE) were fabricated by casting an electroactive material supplemented using PVB binder onto a SS with a surface area of 1 cm². Although PVB is not the most commonly used binder for such applications, it was herein used, due to its excellent adhesive properties and flexibility. Its unique characteristics, such as high binding strength and compatibility with various active materials, make it suitable for the present application [35, 36]. Before fabrication, SS plates were polished using emery paper, ultrasonically cleaned in acetone for 20 min. A suspension of PVB in 1.1 mL THF was stirred for 5 min, followed by graphite addition, which was stirred for an additional 2 h.

Subsequently, 5, 10 or 15 wt% PIN/WC-based NC were added to the suspension, and the mixture was stirred for another 2 h, before being ultrasonicated for 30 min. The prepared suspension was uniformly applied to SS plates using a micropipette, as illustrated in Fig. 1. Coated substrates were dried at 40°C, under 400 mm Hg, for 48 h, left undisturbed for 3 days, and then subjected to further drying at 50 °C, for another 48 h. The electrodes were designated as PIN/WC-based NC I, II and III. This protocol ensured the formation of a uniform and adherent electrode layer, suitable for subsequent analysis.

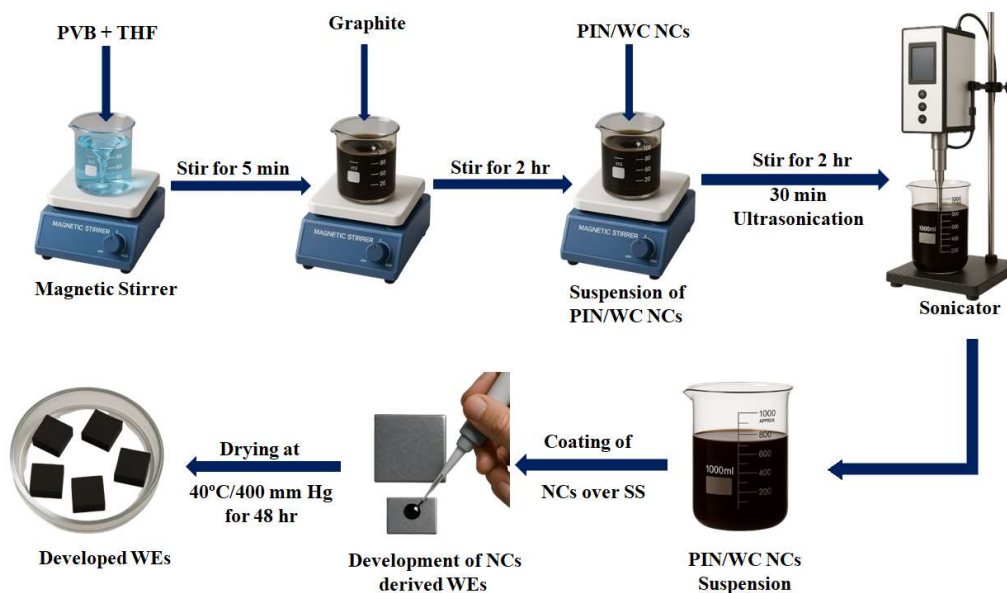


Figure 1: Fabrication of PIN/WC-based NC with enhanced σ DC by modifying PIN matrix with WC.

HE experiments

HE on WE were conducted, over 6 to 18 h, in a 35 L incubator procured from New Brunswick Instruments, Germany. In order to maintain 40% relative humidity, the incubator was charged with 0.5 L deionized water and maintained at 40 ± 1 °C, in the presence of 0.2% CO₂.

Characterization

FTIR spectra were recorded on a Thermo Nicolet spectrometer in KBr, in the range from 4000 to 500 cm⁻¹, in transmission mode. SEM images of WE were captured using a JSM-6610LV (JEOL), at 10 kV, with scales from 5 to 10 μm. EDX analyses were also conducted using a JEOL JSM-6610LV instrument operated at a beam voltage of 15 kV. Since the surface features were distinctly observable, gold coating was not required. UV-vis spectra were obtained using a PerkinElmer Lambda 365 in the wavelength range from 200 to 800 nm. Optical band gaps (E_g) from PIN/WC-based NC were evaluated using Tauc equation:

$$\alpha h\nu = K(h\nu - E_g)^{1/2} \quad (1)$$

by converting diffuse reflectance spectra into Kubelka-Munk (K-M) function. E_g was determined by plotting $(F(R) \times hv)^{1/2}$ as function of 'hv', and then extrapolating the linear part to 'hv' axis [37]. σ_{DC} of WE were measured using a Keithley four-probe nanovoltmeter (2182 A), with a current source (6221 DC) [38]. Simultaneous thermogravimetric analysis (TGA), differential thermogravimetric analysis (DTA) and differential thermal analysis (DTG) were conducted using EXSTAR TGA/DTA/6300 instrument, at a heating rate of 10 °C/min, with a sample size from 10.51 to 10.53 mg, under an airflow of 200 mL/min, across a temperature range from 35 to 700 °C. Thermal stability of PIN/WC-based NC was expressed in terms of thermogravimetric onset (TG_o), thermogravimetric endset (TG_e), peak temperatures of DTA (°C), DTG (°C), rate of degradation (mg/min) and heat of fusion (ΔH_f ; mVs/mg). XRD patterns were obtained using a Rigaku-Geigerflex X-ray diffractometer, with Cu-K α radiation ($\lambda = 1.54$ nm), scanned at a rate of 10°/min, over a 2θ range from 10 to 80°. Crystalline size was determined using Debye-Scherrer equation:

$$D = K\lambda / \beta \cos\theta \quad (2)$$

where $K = 0.9$ is shape factor, λ is X-ray wavelength, β is full width at half maximum (FWHM) of diffraction peak and θ is Bragg angle [39].

Result and discussion

Characterization

FTIR

Fig. 2(a–e) shows FTIR spectra of PIN/ and PING/WC-based NC. PIN spectrum showed peaks at 3138 and 1454 cm^{-1} , which represents N-H and C-H stretching [40, 41]. A peak at 1617 cm^{-1} indicates O-H bending, suggesting the presence of residual water molecules [42] (Fig. 2a). Peaks at 1110 and 1334 cm^{-1} correspond to C-N stretching and bending vibrations [38, 43]. The peak at 1568 cm^{-1} represents deformation of indole moiety [41]. Additionally, the peak at 2852 cm^{-1} corresponds to symmetric C-H stretching in CTAB methylene (–CH₂) groups, influenced by its hydrophobic interaction with PIN [44] (Fig. 2a). WC spectrum shows a significant peak at 665.82 cm^{-1} , which represents W–C's stretching vibration. Wavenumber at 3740.10 cm^{-1} is due to moisture contamination in WC [45] (Fig. 2b). PING peaks at 2922 cm^{-1} were due to C–H's symmetric stretching of graphite. Wave numbers at 741 and 1130 cm^{-1} were due to deformation indole's ring and oxidation. Additionally, PING peaks at 3402 and 1616 cm^{-1} correspond to N-H and O-H stretching vibrations. Peaks at 2855 and 2922 cm^{-1} indicate 2,6-disubstituted derivative of indole. Peaks at 1130, 1245 and 1382 cm^{-1} indicate symmetric vibration of C-C, C-O and C-N bands (Fig. 2c). FTIR spectrum of PIN/WC-based NC III exhibits characteristic stretching vibrations from the aromatic ring, with peaks of C=C (1617 cm^{-1}), C–H (2920 cm^{-1}) and C–C bending (1244 cm^{-1}) (Fig. 2d). Interaction between WC and PIN in NC is confirmed by shifts of PIN peaks to lower wavenumbers, mainly at 1500–1600 cm^{-1} for C=C stretching and 3050 cm^{-1}

for C-H stretching regions. In PIN/WC-based NC III, additional shifts from 1100 to 1600 cm^{-1} indicate changes in benzene ring vibrations, including peaks at 1104 C-N and 1516 cm^{-1} C=C stretching [46]. These spectral shifts highlight vibrational modifications caused by WC-PIN interaction. Furthermore, N-H peak from PIN/WC-based NC III shifts to a lower band (3356 cm^{-1}), after 6 h, at 50 °C (Fig.2e).

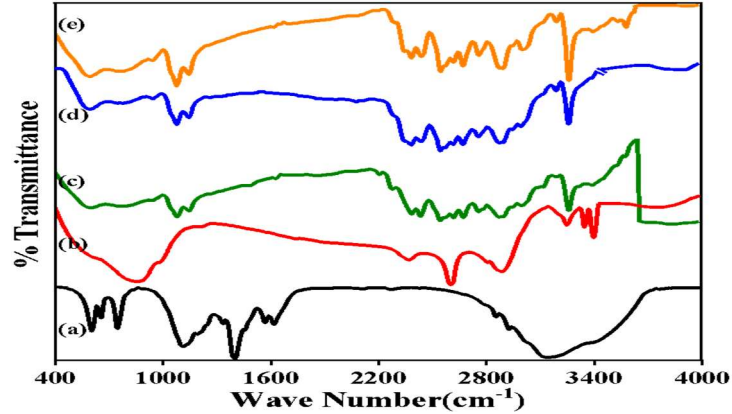


Figure 2: FTIR spectra of PIN (a), WC (b), PING (c), PIN/WC-based NCIII before HE (d) and PIN/WC-based NCIII after HE (e).

Scanning electron microscopy

Fig. 3 illustrates HE's effect on variations in the microstructure from PIN/WC-based NC.

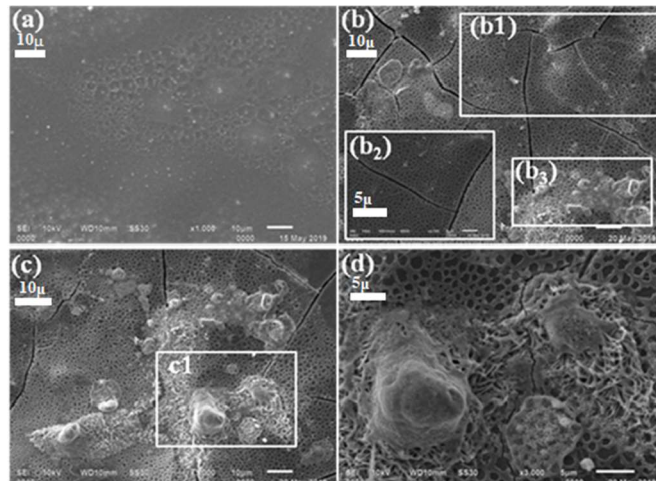


Figure 3: SEM image of PIN/WC-based NC III-(a), (b), (c) before HE and (d) After HE, with (b) occasional cracks, at (b₁) 1,000 magnifications, (b₂) 2,700 magnification, (b₃) phase separation, (c) phase separation at 1,000 magnification, (c₁) occasional cluster of WC and (d) organized cluster of WC, at 3,000X magnification and 5 μm scale.

In order to have comparable results, all micrographs were scanned under identical conditions. Prior HE, WE displayed homogeneous distribution of

PIN/WC-based NC, with occasional porous morphology due to solvent's evaporation (Fig. 3a). However, such WE's porous morphology had no remarkable effect on their σ DC. Exposure under HE has changed WE's morphology into occasional cracks (Fig. 3b) and phase separation (Fig. 3c). In order to have further insight into morphological changes associated with WE under HE, they were examined under high magnifications, wherein well-defined cracks were found over their surface (Fig. 3b₁). Examination of phase separated morphology (Fig. 3b₃), under higher magnification, reveals a well-organized biphasic composition of PIN/WC-based NC (Fig. 3c), with occasional clusters of WC (Fig. 3c₁), which were clearly visible under higher magnifications (Fig. 3d). Microscopic examination clearly indicates that WE's morphology were greatly tarnished on HE derived from PIN/WC-based NC III within 6 h. PIN/WC-based NC III's surface morphology shows some cracks after 6 h exposure to HE, as well as occasional and well organized clusters of WC. To support SEM analysis, EDX was employed to perform qualitative and semi-quantitative elemental analysis. Both PIN and PIN/WC-based NC were examined using EDS, to obtain detailed information about their elemental composition of carbon (C), nitrogen (N), oxygen (O) and tungsten (W), as presented in Table 1.

Table 1: EDX analysis for PIN and PIN/WC-based NC I, II and III.

| Elements (%) | PIN | WC | PIN/WC-based NC I | PIN/WC- based NC II | PIN/WC- based NC III |
|--------------|-------|-------|-------------------|---------------------|----------------------|
| C | 81.33 | 7.29 | 83.56 | 79.65 | 75.88 |
| N | 11.24 | - | 12.09 | 11.98 | 11.65 |
| O | 5.045 | 0.8 | 2.54 | 3.33 | 3.08 |
| W | - | 91.71 | 4.46 | 8.59 | 12.76 |

Thermal characteristics

TGA-DTA-DTG spectra were used to investigate thermal stability of PIN/WC-based NC. Before first-step decomposition, weight losses in these NC were due to moisture expulsion and residual solvents. PIN revealed single-step decomposition with TG_o, at 514 °C, leaving 52% residue. This progressed to 197.6 μ g/°C, with DTG peak temperature of 559 °C. PIN showed -10.6 mVs/mg Δ H_f, with DTA signal (0.65 mV), at 543 °C. Prior to TG_o, 48% PIN degradation was due to moisture expulsion, polymerisation initiators and untreated monomers [32], which was terminated at 573 °C, leaving 1.9% char residue (S1). With TG_e at 600 °C, PIN decomposition was terminated. Corresponding W_r (%) values, at TG_o and TG_e temperatures, were 514 and 573 °C, respectively (Fig. 4). PIN degradation has been supported through DTA signal of 0.654 mV, at 543 °C. DTG reveals degradation rate of 197.6 μ g/°C, at 559 °C (Fig. 5).

Thermal stability of WC was retained up to 400 °C. Due to WC oxidation at 500 °C, samples weight increased up to 2.8% [47, 30]. WC decomposition was progressed at -199 μ g/°C, and 219.3 μ V DTA signal at 554 °C, respectively. TG_e of WC appeared at 625 °C, with 115.5 wt %. PING shows single step degradation, with TG_o at 400 °C,

leaving W_r (%) of 75.74, and TG_e at 554 °C, leaving W_r (%) of 12.4. DTA and DTG data reveals 302 and 68.9 μV , at 535, 592 and 287.7 $\mu g/^\circ C$, at 537 °C.

PIN/WC-based NC III has shown single step decomposition at 122.5 $\mu g/min$, of which ΔH_f shows single step degradation, with TG_o at 400 °C, leaving W_r (%) of 72.53, and TG_e at 463 °C, leaving W_r (%) of 39.1.

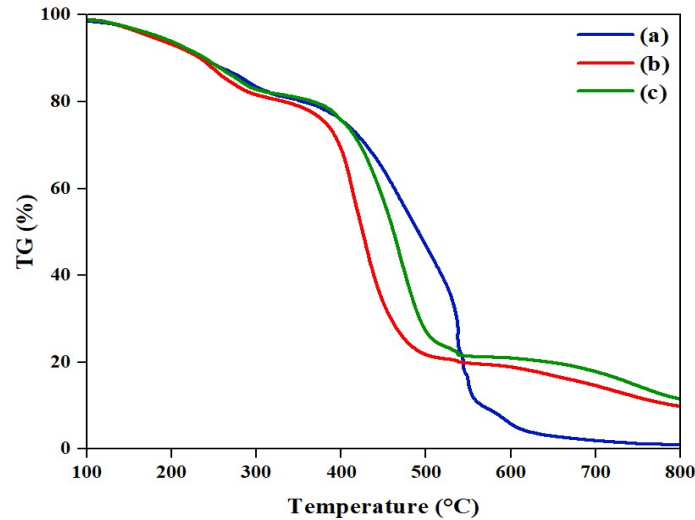


Figure 4: TGA characteristics for (a) PING, (b) PIN/WC-based NC III before HE and (c) PIN/WC-based NC III after HE.

DTA shows 298.7 and 315.4 mV, at 445 and 458 °C, respectively. DTG reveals decomposition of 340.2 and 0.080 mg/ °C, at 465 and 446 °C, respectively. PIN/WC-based NC III HE exhibits degradation of 76.42 and 22.1, at 400 and 516 °C, respectively. DTA and DTG show peaks at 260.9 and 122.5 μV , at 491 and 505 °C (Fig. 5).

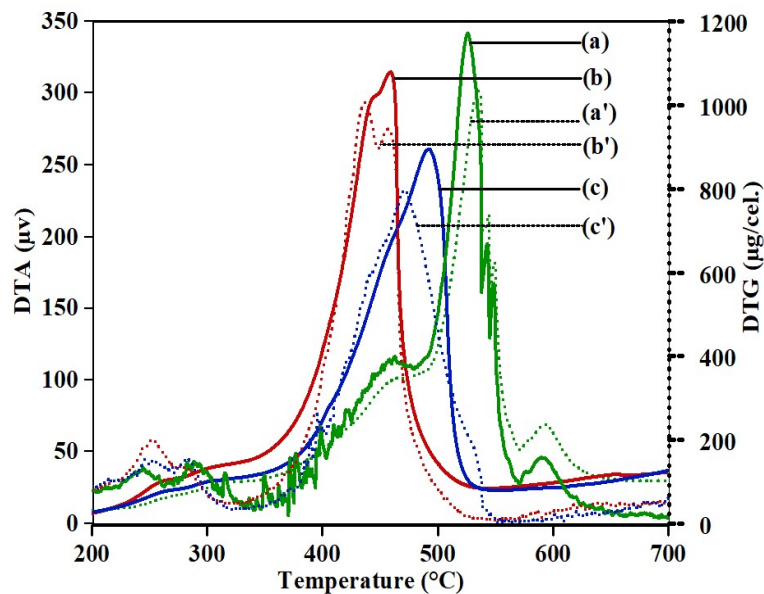


Figure 5: DTA scan of (a) PING; PIN/WC-based NC III (b) before HE and (c) after HE; (a') DTG scan of PING; PIN/WC-based NCIII (b') before HE and (c') after HE.

Optical characterization

UV-vis spectra from PIN, WC and PIN/WC-based NC I, II and III are displayed in Fig. 6. PIN showed well known characteristic peak from 270 to 363 nm, due to $\pi-\pi^*$ and $n-\pi^*$ transitions. Specific peaks at 270 and 306 nm belong to $\pi-\pi^*$ transition of polymer chain and TO conjugated $\pi-\pi^*$ transition of benzene ring, respectively [48, 49].

Diffuse absorbance graph (Fig. 6A) shows that strongest absorption peak for all PIN/WC-based NC occurs at low wavelengths, specifically, at 205 nm. Similarly, peaks for PIN and WC are observed at 204 and 205 nm, respectively. Among samples, PIN/WC-based NCIII exhibits highest absorbance, while WC shows the lowest. An increase in WC concentration within the polymer matrix enhanced peak intensity, indicating strong filler-matrix interactions. Additionally, absorption edge of PIN/WC-based NC shifted slightly toward higher wavelengths, suggesting the development of intermolecular interactions. The % reflectance vs. wavelength spectra (Fig. 6B) complement absorbance data, highlighting the optical behaviour of PIN, WC and PIN/WC-based NC. All samples exhibit low reflectance in UV region (below 300 nm), confirming efficient light absorption. Notably, PIN/WC-based NC III has lowest reflectance, corresponding to its highest absorbance. Optical E_g was estimated with the help of Tauc plot between $(\alpha h\nu)^{1/2}$ and $h\nu$ (Table 2). E_g from PIN, WC, PIN/WC-based NC I, II and III are 2.89 [50], 3.38 [51], 2.75, 2.56 and 2.49, respectively. E_g from PIN/ WC-based III was lowest, which indicates that PIN/WC-based NCIII needs small amount of energy for the transition from valence to conduction band. Above data of E_g values reflect semiconducting nature from PIN/WC and PIN/WC-based NC.

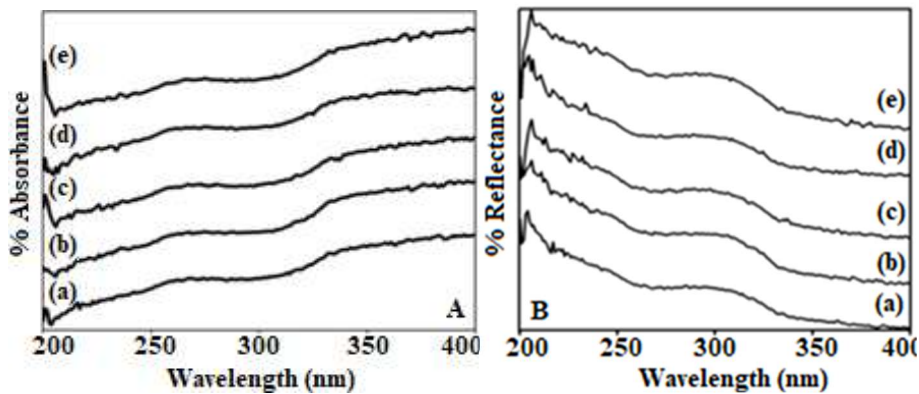


Figure 6: Optical characterization- (A) absorbance and (B) reflectance of (a) PIN, (b) WC and PIN/WC-based NC (c) I, (d) II and (e) III.

Table 2: E_g from PIN, WC and PIN/WC-based NC I, II and III.

| S. no | Samples | E_g (eV*) | References |
|-------|---------------------|-------------|---------------|
| 1 | PIN | 2.89 | [50] |
| 2 | WC | 3.38 | [51] |
| 3 | PIN/WC-based NC I | 2.75 | Present study |
| 4 | PIN/WC-based NC II | 2.56 | Present study |
| 5 | PIN/WC-based NC III | 2.49 | Present study |

XRD analysis

XRD analysis was employed to investigate crystalline structure and phase purity of samples. Diffraction planes for WC and PIN/WC-based NC III are similar, with slighter lower 2θ for the latter. Characteristic diffraction planes at (001), (100), (101), (110), (111), (200), (102) and (201) correspond to hexagonal structure of WC (JCPDS 73–0471), as shown in Fig. 7(a) [52, 53]. Broad diffraction pattern of synthesized PIN in Fig. 7(b) highlights its amorphous nature [54]. Diffraction peaks of PIN/WC-based NC III (Fig. 7c) clearly reveal signature peaks of WC, confirming its successful integration into PIN matrix. Crystallite sizes for WC and PIN/WC-based NC III, calculated using Debye–Scherrer formula, were determined to be 22 and 17 nm, respectively. Reduction in crystallite size (Cs) enhances access to active sites on the WE, leading to a higher Cs value for PIN/WC-based NCIII than that from WC [55].

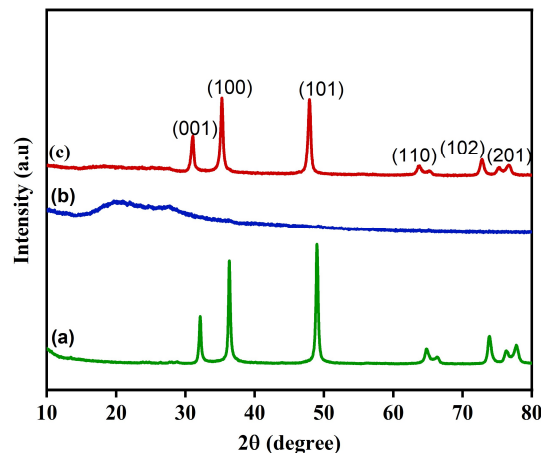


Figure 7: XRD of (a) WC, (b) PIN and (c) PIN/WC-based NC III.

Electrical conductivity

I-V characteristics of WE at room temperature show linear variation, which implies ohmic behaviour of PIN/WC-based NC, as shown in Fig. 8[5].

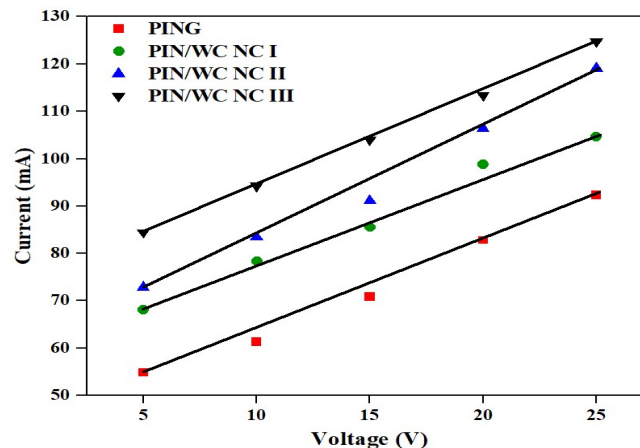


Figure 8: I-V characteristics of PING, PIN/WC-based NC I, II and III coated WE at RT.

The instrument was equipped to measure σ_{DC} values at selected voltages, specifically at 1, 10 and 100 V. To evaluate WE's relative electric behaviour, σ_{DC} was analysed across voltage range from 1 to 100 V, as illustrated in Fig. 9 (a-b). Increased conductivity of PIN/WC-based NC was due to WC's high conductivity and to the favourable interaction between PIN and WC, which enhanced charge transfer process [43]. Conductivity of PIN/WC-based NC increased with higher WC concentrations, particularly at 100 V. Effect of HE on σ_{DC} of WE was studied across various voltages (1–100 V). PIN/WC-based NCIII exhibited maximum conductivity, achieving 25.5×10^{-3} S/cm, at 100 V, at room temperature. Under relative humidity of 40%, PIN/WC-based NC treated for 6, 12 and 18 h showed an increasing trend in σ_{DC} up to 6 h, beyond which conductivity began to decrease. Highest conductivity under HE was also observed in PIN/WC-based NC III, with a value of 36.4×10^{-3} S/cm, at 100 V, as shown in Fig. 9(b) [43, 32].

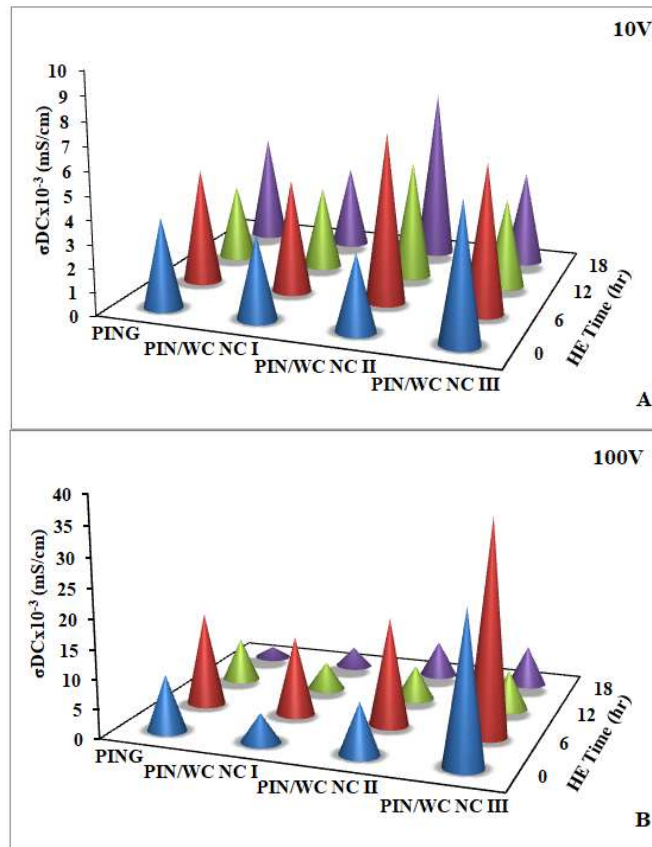


Figure 9: Effect HE on σ_{DC} conductivity over time (h) from PING, PIN/WC-based NC I, II and III, at (A) 10 and (B) 100V.

Comparison of various reported PIN, WC-based NC for thermal, electrical and humidity behaviour

Table 3 provides a comprehensive analysis of reported PIN and WC-based NC synthesized through various techniques, highlighting key parameters such as σ_{DC} , thermal stability and humidity response. Reported NC include PVA/Mn₂O₃-rGO

[56], PIN/PVA/Fe₃O₄ [57], PIN-ALMCM-41 [58], PPy/WC [59], PIN based fibres [60], PVP/CMC/WO₃ [61] and PIN/CNT [62]. Data highlight σ_{DC} and thermal stability (degradation temperature) achieved through diverse metal combinations and synthesis techniques. However, most existing studies have not explored humidity behaviour. This study addresses that research gap, by investigating humidity response of PIN/WC-based NC. These findings emphasize the adaptability of PIN/WC-based NC in optimizing properties for targeted applications.

Table 3: Comparative analysis of PIN and WC-based NC for thermal, electrical and HE.

| S. no | NC | Synthesis method | σ_{DC} (S/cm) | Degradation temperature (°C) | References |
|-------|---|-----------------------------------|---|------------------------------|---------------|
| 1 | PVA/Mn ₂ O ₃ /rGO | Solution casting | 2.023×10^{-8} - 1.895×10^{-5} | 300 | [56] |
| 2 | PIN/PVA/Fe ₃ O ₄ | <i>In situ</i> polymerization | 8×10^{-6} | 183 | [57] |
| 3 | PIN/ALMCM-41 | <i>In situ</i> polymerization | 4×10^{-3} | 599 | [58] |
| 4 | PPy/WC | <i>In situ</i> polymerization | 10^{-4} - 10^{-3} | 350 | [59] |
| 5 | PIN based fibers | Chemical oxidative polymerization | 1.2×10^{-3} | 290 | [60] |
| 6 | PVP/CMC/WO ₃ | Pulsed laser ablation | 1×10^{-8} | 400 | [61] |
| 7 | PIN/CNT | Electro spinning | 10^{-4} - 10^{-3} | 320 | [62] |
| 8 | PIN/WC | Chemical oxidative polymerization | 25.5 - 36.4×10^{-3} | 400 | Present study |

Conclusion

A series of PIN/WC-based NC was herein developed through iron trichloride-initiated chemical oxidative polymerization of indole in CTAB presence, at WC concentrations ranging from 5 to 15 (wt%). FTIR spectra reveal formation of PIN/WC-based NC, along with moisture contamination from WE. Simultaneous TG-DTA-DTG analysis demonstrated enhanced thermal stability of WE up to 6 h of HE. While HE has affected WE's morphology, their σ_{DC} was retained for up to 6 h. WE's σ_{DC} exhibited ohmic behaviour, and increased linearly with its concentration. UV-vis data showed a regular decrease in E_g , with increasing WC concentration.

Among developed composites, PIN/WC-based III demonstrated highest σ_{DC} of 25.5×10^{-3} S/cm (before HE), and a low E_g of 2.49 eV, indicating their excellent potential for applications in semiconducting, energy production and storage devices. Future research could explore additional domains where these PIN/WC-based NC may have significant impacts, such as in advanced optoelectronic systems, environmental sensing technologies and bio-compatible conductive materials. These directions may open new pathways for enhancing applicability and functionality of developed composites in emerging technologies.

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Authors' contributions

B. Arya: investigation; methodology development; original draft preparation. **J. Maheshwari:** data visualization; review and manuscript editing. **A. Bughani and I. Joshi:** formal analysis and editing. **M. G. H. Zaidi:** supervision; resources; aided in writing and editing the manuscript. **S. Mehtab:** supervision; visualization support; aided in writing, reviewing and editing the manuscript.

Abbreviations

AIMCM-41: aluminosilicate material
CMC: carboxy methyl cellulose
CNT: carbon nanotubes
CTAB: cetyltrimethylammonium bromide
DTA: differential thermogravimetric analysis
DTG: differential thermal analysis
ECP: electrical conducting polymers
EDX: energy dispersive X-ray spectroscopy
E_g: optical band gap (eV)
Fe₃O₄: magnetite
FeCl₃: ferric chloride
FTIR: Fourier transform infrared
HE: humidity exposure
Mn₂O₃: manganese(III) oxide
NC: nanocomposites
PANi: polyaniline
PIN: polyindole
PING: polyindolegrafite
PPy: polypyrrole
PVA: polyvinylalcohol
PVB: poly vinylbutyral
PVP: polyvinylpyrrolidone
rGO: reduced graphene oxide
SEM: scanning electron microscopy
SS: stainless steel
TG_e: thermogravimetricendset
TG_o: thermogravimetric onset
TGA: thermogravimetric analysis
THF: tetrahydrofuran
TMC: transition-metal carbide

WC: tungsten carbide
WE: working electrode
WO₃: tungsten trioxide
Wr%: weight residue
wt%: weight percentage
XRD: X-ray diffraction

Symbols definition

ΔH_f: Heat of fusion
σ_{DC}: electrical conductivity

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